Polyurethane Anionomers Using Phenolphthalins. II

S. RAMESH and GANGA RADHAKRISHNAN*

Polymer Division, Central Leather Research Institute, Adyar, Madras 600 020, India

SYNOPSIS

Polyurethane anionomers based on phenolphthalins have been characterized by differential scanning calorimetry, swelling, and mechanical studies. DSC study shows that introduction of ionic groups into the polymer chain lowers the T_g of the polyurethane. Swelling studies reveal that water uptake is high for ionomers. From the tensile data we observe that Zn^{+2} neutralized ionomers showed enhanced properties. © 1996 John Wiley & Sons, Inc.

INTRODUCTION

Polymers based on aromatic units are of interest due to their superior thermal stability and improved mechanical properties like good strength and toughness. Incorporation of aromatic units as chain extenders have been of interest to prepare cast, functionalized, and liquid crystalline polyurethanes.¹⁻³ Morgan's detailed work in preparing polyesters using phenolphthalein and substituted phenolphthaleins⁴ threw a new light on the use of these compounds for polymer synthesis. Subsequently, phenolphthalein and its derivatives have been used to prepare a variety of polymers.⁵⁻⁷

Phenolphthalins, due to their ionic functionality and triphenyl methane units, yield polymers with superior properties. They have been used to prepare heat-resistant polyethers,⁸ fire-resistant thermosets, and thermoplastics for aircraft and aerospace applications.^{9,10} Recently, phenolphthalins and their derivatives have been used to prepare fluorescent poly(arylene ether)s,¹¹ poly(imidoaryl ether ketone)s, and poly(imidoaryl ether sulfone)s.^{12,13}

Polyurethane ionomers are versatile materials that offer an interesting combination of properties to the needs of end user by the incorporation of different ionic groups into the polyurethane chain.¹⁴⁻²⁶ In this work we have used these compounds for the synthesis of polyurethane ionomers and studied their properties.

EXPERIMENTAL

Polyurethanes were prepared by chain extending the prepolymer based on PTMG and TDI with the phthalins in different mole ratios. They were then converted into sodium and zinc anionomers. The detailed synthetic procedure is given in our earlier publication.²⁷ The polymer and ionomer codes are given in Table I.

Physicochemical Characterization Techniques

Differential scanning calorimetric study was performed using a Du Pont 910 instrument equipped with a standard cell at a heating rate of 10° C/min under nitrogen atmosphere. The scanning temperature range was from -100 to $+120^{\circ}$ C. The samples were quenched on the cell using liquid nitrogen to attain low temperatures. The amount of samples taken for each run was 5 ± 0.2 mg.

Swelling studies were performed for the samples in water. Samples of size 2×2 cm were cut and immersed in water at 30°C. The samples were taken out at stipulated times, wiped with a tissue paper, and allowed to stand for a minute in air and then weighed accurately.

Mechanical properties of the polymers were evaluated using an Instron universal testing machine model 4501 at a crosshead speed of 50 mm/min. The specimens were cut into a dumbbell shape of length 2.1 cm and breadth 0.4 cm. An average of four specimens for a sample was considered.

^{*} To whom correspondence should be addressed. Journal of Applied Polymer Science, Vol. 60, 2273–2277 (1996) © 1996 John Wiley & Sons, Inc. CCC 0021-8995/96/122273-05

Polymer Compositions	Polymer Codes	T_{g} (°C)	Tensile Strength at Break (MPa)	% Elongation at Break
$PTMG_{1000}/2TDI/Phenolphthalin$	PH	-55.0	2.50	1293
	PH-Na	-61.0	4.70	2869
	PH-Zn	-58.0	12.10	2004
PTMG ₁₀₀₀ /2TDI/Phenolphthalein	PH_{R}	-59.5	0.53	502
$PTMG_{1000}/2TDI/Thymolphthalin$	тн	-46.0	0.39	2020
	TH-Na	-52.5	0.62	2505
	TH-Zn	-50.0	1.33	1208
$PTMG_{1000}/2TDI/Tetrabromo phenolphthalin$	BrPH	-50.5	0.17	189
	BrPH-Na	-54.5	0.34	467
	BrPH-Zn	-53.0	0.98	296
PTMG ₁₀₀₀ /2TDI/Phenolphthalin + 1,4-butane diol	PHB	-50.0	0.38	763
	PHB-Na	-55.5	0.53	1347
	PHB-Zn	-53.5	1.05	986

 Table I Glass Transition Temperatures and Mechanical Properties of the Polyurethanes and their

 Anionomers

RESULTS AND DISCUSSION

DSC Study

The DSC curves of polyurethanes and that of ionomers are given in Figures 1–3. The midpoint of the transition is given as the glass transition temperature and the values are mentioned in Table I. For all the polyurethanes the T_g was well below the room temperature, which is characteristic of PTMG₁₀₀₀-based polymers. For these chain extended



Figure 1 DSC curves of polyurethanes.

systems the transitions of the hard segments were not clearly observable. This is consistent with the work by Cooper et al., based on the work on the polyurethane anionomers using dimethylol propionic acid.¹⁶ For the PHB polymer, a broad endotherm observed at 41°C is due to a short range ordering of hard domains. This is due to the introduction of butanediol units along with phenolphthalin in the chain extension stage.

When we compare the T_g of the ionomer to those with carboxylic acid groups, we observe a shift in T_g towards lower values. This is due to phase separation caused by the realignment of hydrogen bonding between the hard and soft segments by the ionic groups



Figure 2 DSC curves of polyurethane ionomers with Na^+ counterions.



Figure 3 DSC curves of polyurethane ionomers with Zn^{2+} counterions.

leading to more phase purity of soft segments. Thus, the T_g shifts towards that of the basic polyol unit. When we compare the T_g values of polyurethanes with Na and Zn as counterions, no appreciable shift is observed, which is also consistent with the results observed by Cooper et al.¹⁶ But the magnitude of phase separation observed here for the ionomers is less than those of carboxylate systems based on aliphatic groups. This may be due to the rigidity of these aromatic systems, which hinder significant phase separation of the hard segments, which are dispersed well in the soft segments.

Swelling Studies

The water absorption studies of polyurethanes and ionomers are shown in Figures 4 and 5. Generally, ionomers showed increased water absorption than their nonionic counterpart. The water uptake value of PH and PH_R were low when compared to PH-Na and PH-Zn. Once the carboxyl groups were converted into the sodium salt, water uptake increases. When we compare the ionomers containing sodium as a counterion with that of zinc as a counterion, water uptake is more for the former. This is due to chemical crosslinking by the divalent metal ion zinc, which is also proved by mechanical properties study.

Mechanical Properties

The uniaxial stress-strain graphs of the polyurethane and ionomers are given in Figures 6 and 7. The ultimate tensile strength and percentage of elongation at break are given in Table I. Polyurethanes containing -COOH groups had enhanced properties compared to the reference sample PH_{R} without the -COOH groups. All the ionomers showed an increase in tensile strength more than their base polymer. As the ionic groups are introduced into the polymer chain the aggregated hard segments act as strong physical networks, leading to an increase in tensile strength. Among the polyurethane ionomers, those with zinc as counterions showed enhanced tensile strength. This is due to the chemical crosslinking of Zn⁺² ions across two carboxyl groups apart from the columbic forces. This ionic crosslinking imparts more chain stiffness to



Figure 4 Water absorption behavior of (\times) PH_R, (\blacksquare) PH, (\bullet) PH-Na, (\blacktriangle) PH-Zn, (\Box) TH, (\bigcirc) Th-Na, and (\triangle) TH-Zn.



Figure 5 Water absorption behavior of (\Box) BrPH, (\bigcirc) BrPH-Na, (\triangle) BrPH-Zn, (\blacksquare) PHB, (\bullet) PHB-Na, and (\blacktriangle) PHB-Zn.

the polyure thane chain and, thus, elongation is less than that of $\rm Na^+$ as a counterion.

The overall excellent elongation observed for these polymers is due to the stress-induced crystallization of the soft segment, which stops early failure of the sample. Ionization of the -COOH groups drives the system towards more phase separation, which in turn, enhances the ability of the soft segment to crystallize under further strain. When we



Figure 6 Stress-strain curves of (a) PH systems and (b) TH systems.



Figure 7 Stress-strain curves of (a) BrPH systems and (b) PHB systems.

trabromo phenolphthalin or thymolphthalin (TH and BrPH). One reason may be due to the reasonably higher molecular weight of the former.²⁷ Apart from this, phenolphthalin moieties introduce rigid aromatic hard segments imparting good mechanical strength. In the case of substituted phenolphthalins as chain extenders, efficient packing of the phthalin units in the chain is hindered due to the alkyl and halogen groups on the phenyl rings. In the case of the PHB polymer using a mixed chain extender (phenolphthalin and butanediol) ordered arrangement is hindered due to a difference in the size and shape of the extenders. Thus, PH possess more superior properties than PHB. Chain extended polyurethanes synthesized using phenolphthalins, and the prepolymer based on PTMG units have been found to be of extensive application in the area of coatings onto leather, paper, and textiles.

CONCLUSIONS

Differential scanning calorimetric study shows that the phase separation occurs in ionomers, leading to more soft segment purity. Swelling studies reveal that water uptake is higher for ionomers than their nonionic counterpart. Tensile strength data indicate that polyurethanes containing Zn^{+2} ions possess enhanced mechanical properties.

One of the authors (S. Ramesh) would like to thank the Council of Scientific and Industrial Research (CSIR), India for a fellowship.

REFERENCES

- 1. S. Ramesh, P. Rajalingam, and G. Radhakrishnan, Polym. Int., 25, 253 (1991).
- C. X. Lu, Y. Yang, C. Xiao, and A. Ji, J. Polym. Sci., Polym. Chem. Ed., 25, 3267 (1987).
- K. R. Haridas and G. Radhakrishnan, J. Polym. Sci., Polym. Chem. Ed., 33, 901 (1995).
- P. W. Morgan, J. Polym. Sci., Polym. Chem. Ed., 2, 437 (1964).

- G. Quipeng, H. Jinyu, and C. Tianlu, *Polym. Commun.*, 31, 115 (1990).
- G. Quipeng, H. Jinyu, and C. Tianlu, J. Appl. Polym. Sci., 42, 2851 (1991).
- H. Rajan, P. Rajalingam, and G. Radhakrishnan, J. Appl. Polym. Sci., 48, 2095 (1993).
- 8. Jpn. Kokai Tokkyo Koh. JP 02,318,720 CA 114: P82793v.
- 9. Eur. Pat. Appl. EP 336420.
- A. A. Kourtides, Pap. Fires Retard Chem. Assoc., 34– 82, 1978 (Eng). CA: 93-73105t.
- S. Matsuo, N. Yakoh, S. Chiuo, M. Mitani, and S. Tagami, J. Polym. Sci., Polym. Chem. Ed., 32, 1071 (1994).
- M. Struklej and A. S. Hay, *Macromolecules*, **24**, 6870 (1991).
- M. Struklej and A. S. Hay, *Macromolecules*, **25**, 4721 (1992).
- D. Dietrich, W. Keberle, and H. Witt, Angew. Chem. Int. Ed., 9(1), 40 (1970).
- J. A. Miller, K. K. S. Hwang, and S. L. Cooper, J. Macromol. Sci. Phys. Ed., B22, 321 (1983).
- C. Z. Yang, T. G. Grasel, J. L. Bell, R. A. Register, and S. L. Cooper, J. Polym. Sci., Polym. Phys. Ed., 29, 581 (1991).
- 17. T. Y. T. Chui, P. K. H. Lam, C. Butler, M. H. George, and J. A. Barrie, *Polym. Commun.*, **29**, 317 (1988).
- P. K. H. Lam, M. H. George, and J. A. Barrie, *Polym. Commun.*, **30**, 2321 (1989).
- P. K. H. Lam, M. H. George, and J. A. Barrie, *Polym. Commun.*, **32**, 80 (1991).
- S. Ramesh and G. Radhakrishnan, J. Macromol. Sci., Macromol. Rep., A30 (Supp.3-4), 251 (1993).
- S. Ramesh and G. Radhakrishnan, J. Macromol. Sci., Macromol. Rep., A32(1&2), 91 (1995).
- 22. H. Rajan, P. Rajalingam, and G. Radhakrishnan, Polym. Commun., **32**(3), 93 (1991).
- H. A. Al-Salah, K. C. Frisch, X. H. Xiao, and J. A. Mclean, Jr., J. Polym. Sci., Polym. Chem. Ed., 25, 2127 (1987).
- S. Ramesh and G. Radhakrishnan, J. Polym. Mater., 12, 71.
- B. K. Kim and T. K. Kim, J. Appl. Polym. Sci., 43, 393 (1991).
- S. A. Chen and J. S. Hsu, Makromol. Chem., 193, 423 (1992).
- S. Ramesh and G. Radhakrishan, *Polymer*, **35**, 3107 (1994).

Accepted January 10, 1996